Aluminum Composite Materials Providing Extended Service Life

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Abstract: Hardness and tribotechnical characteristics were investigated under conditions of dry sliding friction on steel obtained by liquid-phase mixing of Aluminum-Matrix Composite (AMC) materials based on aluminum alloys AK12, D16, V124, AL9, AL25, reinforced with silicon carbide SiC particles with a size of 28 μm with a content of 5, 10 or 15% by volume. AMC performed better than matrix alloys and the commonly used antifriction aluminum alloy AOM 2-1. The dry friction coefficient was 1.5-3 times lower on average, and the volumetric wear rate was 5-9 times lower. An increase in the content of SiC particles in the composite from 5 to 20 vol.%. As a rule, leads to an improvement in the tribotechnical characteristics. The composites obtained have shown a sufficiently high operational suitability for work in friction units with steels both in dry friction and in friction with lubrication.

Keywords: composite materials, wear resistance, aluminum alloys

1. Introduction

Aluminum Matrix Composite (AMC) materials have been found widely applied in mechanical engineering, instrument making, aircraft engineering, and space technology [1]-[8]. An important advantage of AMC is the possibility of their formation for specific operating conditions, due to the introduction of high-strength high-modulus fillers of various types, fractional composition, and volumetric content into a plastic metal matrix. Recently, there has been a tendency to use nanoscale reinforcing phases to improve the properties of AMC, which makes it possible to obtain a more favorable set of characteristics [9]-[16].

The data of laboratory research and industrial tests have shown that AMC can be effectively used in friction units of various types of equipment [12]-[22]. The advantages of aluminum alloys as matrices of such composites are high thermal conductivity, heat capacity, high technological properties, including the possibility of varying mechanical properties by choosing alloying systems. Particles with hardness sharply different from the aluminum matrix are used as reinforcing fillers, which increase the wear resistance, and also reduce and stabilize the friction coefficients in a wide range of loading parameters. An important requirement for such reinforcing phases is high technological compatibility with an aluminum matrix [1]-[3], [16], [21]. The use of carbide particles, including silicon carbide SiC, has shown itself to be quite good [16]-[20]. The most developed technology for obtaining antifriction AMCs are different versions of the method of liquid-phase mixing (introduction of reinforcing particles into the aluminum alloy melt) [14]-[22].
For example, in [15] high-strength aluminum alloy AK12M2MgN (wt.%: 11-13 Si, 1.5-3 Cu, 0.3-0.6 Mn, 0.85-1.35 Mg, < 0.5 Zn, 0.05-1.2 Ti, 0.3-1.3 Ni, < 0.8 Fe, < 0.2 Cr, < 0.1 Sn) were reinforced with particles of α-SiC powder with dimensions of 40 and 14 μm with using the method of mechanical mixing at a temperature of 800-850 °C and an impeller rotation speed of 600 min⁻¹. Under conditions of dry sliding friction according to the “bushing (steel 40X)-disk (AMC)” scheme at a speed of 0.39 m/s at specific loads of 0.2-0.7 MPa, the introduction of 10% SiC reduced the wear rate by 1.8 ... 2.9 times and reduced the coefficient of friction at a load of 0.2 MPa by ≈ 1.2 times. With the addition of 5% SiC, the tribotechnical properties improved to a lesser extent. The bushings of bearings of radial pairs of electric centrifugal pumps of oil refining equipment, made of AMC, have shown themselves well when tested in an aqueous medium and in an environment that simulates formation fluid.

According to [16], AMC obtained by mechanical mixing of 5 wt.% SiC particles 28 μm in size into the melt of aluminum alloys AL2 (wt.%: 84.3-90 Al, 10-13 Si) and D16 (wt.%: 3, 8-4.9 Cu, 1.2-1.8 Mg, 0.3-0.9 Mn), showed in comparison with the matrix material: an increase in the hardness HB from 624 to 712 (for AL2) and from 849 to 1,110 (for D16). Under conditions of dry sliding friction “bushing (steel 40X)-disk (AMC)” at a speed of 200-1,500 min⁻¹ at specific loads of 70-215 N for AMS, as compared to the matrix alloy, the load and the speed of transition to the “scuffing” mode increased accordingly from 70 to 180 N and from 600 to 1,500 rpm (for AL2) and from 1,000 to 1,500 rpm at the same load (for D16). The dry friction coefficient, depending on the sliding speed and load, decreased from 0.24-0.39 to 0.13-0.16 (for AL2) and from 0.18-0.25 to 0.14-0.22 (for D16). At sliding speeds of more than 1,100 min⁻¹, the friction coefficient for AMC based on D16, on the contrary, increased in comparison with the matrix alloy to 0.23-0.26. The wear rate for AMS based on AL2 in comparison with the matrix alloy decreased from 70 to 180 N and, at a speed above 600 min⁻¹, the wear rate, on the contrary, increased by a factor of 1.3-1.35.

In [18], based on the aluminum alloy A359 (wt.%: 9.1 Si, 0.58 Mg), AMC containing 5, 10 and 15 wt.% SiC. The composite melt was compressed in steel molds at 100 MPa. Wear tests were carried out according to the “pin (AMC)-on-disk (steel with a hardness of 62 HRC)” scheme at a speed of 655 min⁻¹ (2.75 m/s) at specific loads of 20, 40, and 60 N. SiC particles were relatively evenly distributed in the matrix alloy. It was found that with increasing SiC content, the wear rate decreases. So the wear rate of AMC with 15 wt.% SiC was 2.5-10 times lower than that of the matrix alloy. The friction coefficient with the addition of SiC decreased only at a load of 20 N. At loads of 40 and 60 N, the friction coefficient was: 0.31-0.32 for the matrix alloy, 0.28-0.3 for AMC with 5 wt.% SiC, for AMS with 10 and 15 wt.% SiC 0.34-0.37. The wear surface of the AMC had a smooth, soft layer of iron oxides.

Researchers [20] obtained AMC samples with 5, 10, and 15 wt.% SiC for engineering applications by the method of stirring casting based on the Al 2014 alloy. In this study, the original base and its composites with 5, 10, and 15 wt.% SiC, made by the method of casting with stirring, were subjected to wear testing. Wear tests of matrix alloy and AMC samples with sliding friction according to the pin-on-disk scheme at a speed of 2 m/s at a load of 20 N showed that AMC with 10 wt.% SiC has the best characteristics. Additionally, samples of such an alloy were extruded. Extrusion improved the tribotechnical characteristics, and the sample extruded in a ratio of 8:1 showed the highest wear resistance.

It should be noted that AMCs based on low-alloy cast alloys, reinforced with SiC particles with sufficiently large sizes of 40-68 microns, were mainly investigated. Wear tests were mainly carried out in a narrow range of loads and speeds, and mainly at rather low values. The results on the most favorable (from the point of view of tribotechnical properties) content of SiC particles in AMC samples differ in different works. There is also little data on macrohardness similar to AMC.

In this regard, in this work, we studied AMC obtained by liquid-phase mixing based on fairly representative types of aluminum alloys AK12, D16, V124, AL9, AL25, reinforced with silicon carbide particles SiC 28 μm in size with a content of 5, 10, or 15 vol..% Sliding friction tests were carried out at sufficiently large values and ranges of speed and load. The hardness HB and the reinforcement hardening efficiency were investigated.

2. Materials and methods
2.1 Materials

Standard aluminum alloys AK12, D16, W124, AL9, AL25 were used as materials for the metal matrix. Their
For the reinforcement, we used a standard commercial SiC powder with particles about 28 μm in diameter, having a hardness of HV = 24 ... 28 GPa and an elastic modulus of 350-490 GPa. SiC powder was added in an amount of 5, 10, or 15% by volume. With a higher content of the reinforcing component, the deterioration of the mechanical properties obtained by AMC is possible [2], [23]-[26]. Billets in the form of castings were obtained by the method of liquid-phase mechanical mixing [2], [14]-[16]. SiC particles heated to about 580 °C were fed into a bath with a melt having a temperature of about 820 °C. Mixing was carried out with an impeller speed of 600 rpm for 10 minutes. In some cases, when obtaining model parts, graphite particles with a size of about 400 μm in an amount of 2.5 vol.% were also kneaded into the melt. To obtain prototypes and model parts, the castings were subjected to traditional machining on a lathe using a tool with a diamond insert. Samples for comparison of properties were obtained from the castings of the above-mentioned aluminum alloys, in which the reinforcing particles were not introduced.

### 2.2 Research methods

The microstructure of the obtained AMC samples was studied using an Olympus GX51 metallographic light microscope.

The hardness HB of the studied samples was assessed according to the standard Brinell method on an Emco-test "Dura Vision 20" hardness tester with an indenter in the form of a ball with a diameter of D = 2.5 mm at a load of P = 613 N. At least 5 measurements were carried out on each sample. To assess the efficiency of strengthening $E_h$ of the obtained AMCs due to the introduction of SiC particles, the following formula is proposed:

$$E_h = \frac{(HB_C - HB_M)}{\% \text{ SiC}}$$

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**Table 1. Chemical composition and mechanical properties of the investigated aluminum alloys**

<table>
<thead>
<tr>
<th>Alloy</th>
<th>Content of elements in wt.%</th>
<th>Mechanical properties</th>
<th>Other elements</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>Cu</td>
<td>Mg</td>
<td>Mn</td>
</tr>
<tr>
<td>AK12</td>
<td>$\leq 0.6$</td>
<td>$\leq 0.1$</td>
<td>$\leq 0.5$</td>
</tr>
<tr>
<td>D16</td>
<td>3.8-4.9</td>
<td>1.2-1.8</td>
<td>0.3-0.9</td>
</tr>
<tr>
<td>W124</td>
<td>3-4</td>
<td>0.15-0.35</td>
<td>0.1-0.3</td>
</tr>
<tr>
<td>AL9</td>
<td>-</td>
<td>0.2-0.4</td>
<td>-</td>
</tr>
<tr>
<td>AL25</td>
<td>1.5-3</td>
<td>0.8-1.3</td>
<td>0.3-0.6</td>
</tr>
</tbody>
</table>
where HB_C is the hardness of AMC, HB_M is the hardness of the matrix alloy, % SiC is the volumetric content of silicon carbide in the sample. The hardening efficiency Eh shows how much the AMC hardness increased when 1 volume % SiC was added to the material. To assess the uniformity of hardening during mechanical mixing, the distribution of hardness HV over the cross-section of the specimens was assessed by the standard Vickers method using a diamond tetrahedral pyramid as an indenter with an indentation load of 0.098 N and holding the indenter under a load of 30 s. For a sample of each AMC, a series of 10 measurements with a step of 0.5 mm was carried out on a Dura Scan 70 hardness tester. The selected load on the indenter ensured that the size of the diagonal of the indentation was significantly larger than the size of the structural components.

Friction and wear tests were carried out on an MI-1M testing machine of the “Amsler” type according to the “disk-block” loading scheme. Similar test schemes were used in articles [14]-[16], [18]-[20]. A disk made of 40X structural steel (analogue of steels 1.7034, 5140H, 40Cr, 38C4) with a hardness of HRC 45 ... 50 rotated at a speed of \( \eta = 300, 1,000 \) and \( 1,500 \) rpm, and an AMC block was pressed against the end surface of the disk with a constant force \( F \) from 70 to 300 N. The test scheme and the main dimensions of the samples and disk are shown in Figure 1. Steel 40X was used since it is the most common contact material. Counter body materials similar in composition and hardness were used in [15], [16]. The tests were carried out under dry friction conditions and friction conditions lubricated with TM-3-18 machine oil. This oil is a multigrade mineral oil for heavily loaded friction units and meets the following standards: SpiraxEP 90W Mobil, Mobilube GX 90 BP, Gear oil EP SAE 90 Esso. The tribotechnical behavior of the samples was evaluated by the value of the friction coefficient \( f \), the intensity of volumetric wear \( I_V \), and the loss of mass of parts \( \Delta m \). The value of the friction coefficient is calculated by the formula: \( f = \frac{M_f}{r(\tau - \sigma)} \), where \( M_f \) - the friction moment, \( r \) - the disk radius, \( \tau \) - the acting load. Friction tests were carried out following the standards: GOST 27860-88 “Rubbing mating machine parts. Methods of measuring wear”, GOST 27640-88 “Engineering materials and lubricants. Methods of experimental evaluation of friction coefficient”, GOST 23.301-78 “Products wear resistance assurance. Instruments for evaluating wear by the method of cut notches. Technical requirements”. A sample of a typical antifriction alloy based on Al-AOM 2-1 (Al-2% Sn-1% Mg) was used as a reference sample.

3. Results and discussion

3.1 Microstructure research

The microstructures of the studied AMCs obtained using a metallographic microscope have much in common. The most typical microstructures are shown by the example of AMC AK12 + 5% SiC and D16 + 5% SiC in Figure 2.

AMC based on casting alloys AK12, W124, AL9, AL25 have a similar matrix alloy microstructure, similar to the base alloy AK12 (Figure 2a), which is an \( \alpha + Si \) eutectic. In AMC based on alloy D16 (Figure 2b), weakly defined grain boundaries are observed, and SiC particles are located between grains of the matrix alloy. For all studied AMCs, SiC particles have the form of fragments, but a dark rim is observed along with the particle-matrix alloy boundaries, which
contains the reaction products between silicon carbide and an aluminum alloy, which provide a strong bond at the filler-matrix alloy interface and a sufficient level of performance properties.

Figure 2. Typical microstructure of aluminum-matrix composite materials obtained by mechanical kneading into the melt: (a) AK12 + 5% SiC (1-α solid solution, 2-eutectic Al-Si (α + Si), 3-SiC particles, 4-needle-shaped Si particles); (b) D16 + 5% SiC (1-matrix alloy D16, 2-SiC particles). Photographs of microstructures were obtained with a light microscope.

3.2 Hardness test

The results of measuring the hardness of the samples by the Brinell HB method are presented in Table 2. It can be seen that the hardness of AMC in all cases exceeds the hardness of the matrix alloy by a factor of 1.1 ... 1.4. The greatest increase in HB values is observed for samples AMC D16 + 5% SiC and W124 + 15% SiC. The hardening efficiency is on average about 2.5 HB units per addition of 1 vol.% SiC. Sample AMC D16 + 5% SiC stands out in particular, for which a weighted value of the hardening efficiency of 6.2 is observed. The high efficiency of hardening for these specimens is since alloys D16 and W124 initially have significantly higher mechanical characteristics in comparison with the rest of the studied matrix alloys. The data on hardness HB very closely coincide with the data of [16].

<table>
<thead>
<tr>
<th>No</th>
<th>Sample AMC</th>
<th>Matrix alloy hardness HB_{al}</th>
<th>Composite hardness HB_{c}</th>
<th>( E_h )</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>AK12 + 5% SiC</td>
<td>620</td>
<td>710 ± 20</td>
<td>1.8</td>
</tr>
<tr>
<td>2</td>
<td>D16 + 5% SiC</td>
<td>850</td>
<td>1160 ± 30</td>
<td>6.2</td>
</tr>
<tr>
<td>3</td>
<td>W124 + 10% SiC</td>
<td>880</td>
<td>1130 ± 20</td>
<td>2.5</td>
</tr>
<tr>
<td>4</td>
<td>W124 + 15% SiC</td>
<td>880</td>
<td>1240 ± 20</td>
<td>2.4</td>
</tr>
<tr>
<td>5</td>
<td>AL9 + 5% SiC</td>
<td>980</td>
<td>1130 ± 20</td>
<td>3</td>
</tr>
<tr>
<td>6</td>
<td>AL25 + 10% SiC</td>
<td>980</td>
<td>1240 ± 50</td>
<td>2.6</td>
</tr>
</tbody>
</table>

The results of measuring the hardness HV over the cross section of AMC samples are shown in the form of graphs in Figure 3. It can be seen that for all AMCs the distribution of microhardness over the cross-section is fairly uniform,
no sharp peaks are observed. At elevated SiC contents, the distribution is less uniform, there are differences in hardness values depending on the measurement point, which is explained when studying the microstructure studied with a metallographic microscope.

Figure 3. The results of measuring the hardness HV of the samples: (a) at low values of the SiC content, (b) at the average values of the SiC content

3.3 Study of tribotechnical characteristics

The values of the friction coefficients $f$ and the intensity of volumetric wear $I_v$ obtained as a result of the tests for the studied AMC, matrix alloys, and the control antifriction alloy AOM 2-1 are shown in Table 2.

As can be seen from the test results, AMCs have friction coefficients, as a rule, lower in comparison with antifriction alloy AOM 2-1 and corresponding matrix alloys. At the same time, they keep them in a much wider range of sliding speeds and loads. An increase in the SiC content leads to a noticeable decrease in the friction coefficients $f$. For example, for AMC based on D16, an increase in the SiC content from 5 to 20% leads to a threefold decrease in $f$. When compared with the data [16], where other test conditions were somewhat different, the nature of the change in the friction coefficients $f$ for AMC D16 + 5% SiC is quite similar: at a load of 70 N, the values of $f$ are greater than that of the matrix alloy, and at a load of 108 N they become close enough.
Table 3. Results of testing samples for friction without lubrication

<table>
<thead>
<tr>
<th>No</th>
<th>Sample</th>
<th>Load F, N</th>
<th>at sliding speed n, rpm</th>
<th>$I_v$, mm$^3$/m$^2$</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>AK12</td>
<td>70</td>
<td>300</td>
<td>0.5</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>1000</td>
<td>0.34</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>1500</td>
<td>0.68</td>
</tr>
<tr>
<td>2</td>
<td>AK12 + 5% SiC</td>
<td>70</td>
<td>0.5</td>
<td>0.019</td>
</tr>
<tr>
<td>3</td>
<td>AK12 + 10% SiC</td>
<td>70</td>
<td>0.68</td>
<td>0.005</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>144</td>
<td>0.64</td>
</tr>
<tr>
<td>4</td>
<td>AL25</td>
<td>70</td>
<td>0.68</td>
<td>0.054</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>108</td>
<td>0.84</td>
</tr>
<tr>
<td>5</td>
<td>AL25 + 5% SiC</td>
<td>108</td>
<td>1.02</td>
<td>0.054</td>
</tr>
<tr>
<td>6</td>
<td>D16</td>
<td>108</td>
<td>1.02</td>
<td>0.054</td>
</tr>
<tr>
<td>7</td>
<td>D16 + 5% SiC</td>
<td>70</td>
<td>1.21</td>
<td>0.066</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>108</td>
<td>0.61</td>
</tr>
<tr>
<td>8</td>
<td>D16 + 20% SiC</td>
<td>300</td>
<td>0.31</td>
<td>0.027</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>0.27</td>
<td>0.20</td>
</tr>
<tr>
<td>9</td>
<td>AOM 2-1</td>
<td>70</td>
<td>0.70</td>
<td>0.047</td>
</tr>
</tbody>
</table>

$I_v$ determined after tests with a load of 5 kg and n = 180 rpm for 15 minutes

Figure 4 shows the average wear rate for ACM with different SiC content (a) and different matrix alloys (b) to evaluate the contribution made by the properties of the matrix alloy. It can be seen that the intensity of volumetric wear $I_v$ for AMC is 5-9 times lower than for the AOM 2-1 alloy and 2-9 times lower than for the corresponding matrix alloys. An increase in the SiC content in materials leads to a decrease in the $I_v$ value. (Figure 4a). A similar dependence was observed in [18] but under different test conditions and a different matrix alloy. In [20], for AMC with a matrix alloy Al 2014, which is close in composition to the AK12 alloy, a similar picture was also observed: AMC with 10 wt.% SiC had the best wear resistance characteristics. The influence of the composition of the matrix alloy on the value of the volumetric wear is also traced (Figure 3b). Figure 4b shows the average wear rate for AMC with 5% SiC for different matrix alloys to evaluate the contribution made by the properties of the matrix alloy. Smaller values of $I_v$ were observed for AMC based on the AL25 alloy with a higher content of Si, Ni, Cr, and Zn in comparison with the matrix alloys AK12 and D16. At the same time, the lowest values of volumetric wear $I_v$ were shown by AMC D16 + 20% SiC and AK12 + 10% SiC, which have high values of hardness.

Figure 5 shows the values of the mass loss $\Delta m$ for AMC parts, counterbody made of 40X steel, and the total mass loss for both parts under conditions of friction without lubrication and with lubrication. Analysis of the behavior of a friction pair in which one part is made of AMC (light parts of the columns in Figure 5), and the counterpart is made of structural steel (dark parts of the columns in Figure 5), makes it possible to assess the compatibility of these materials based on the value of the total wear of the friction pair, which will demonstrate the functioning of the material in the product, since it is important to evaluate the behavior of not only the material under study but also the material of the counterpart paired with which the operation is supposed to be. In general, all AMCs have shown sufficiently high compatibility in pairs with steel, which shows the serviceability of these AMCs for work in friction units with steel. The most balanced ratio of $\Delta m$ for parts made of AMC and steel, as well as lower values of $\Delta m$ for AMC, were shown by materials 1-AL25 + 5% SiC, 2-D16 + 20% SiC, which have high hardness values. Also, AMC with such matrix alloys, as shown above, are distinguished by a lower wear rate, especially with an AL25 matrix. Taking into account the above-considered dependence of the AMC wear intensity on the SiC content, it is recommended to use composites with a SiC content of about 20% for friction units with a pair of “AMC-steel”.

Figure 4. Influence of various factors on the wear rate of AMC under dry friction conditions:
(a) the effect of the SiC content in the AK12 matrix alloy;
(b) $I_w$ values for AMC with different matrix alloys at the same SiC content of 5 vol.%

Figure 5. Loss of mass of parts $\Delta m$ in grams for parts made of steel 40X and the total loss of mass for both parts along the friction path of 9,432 m under friction conditions: (a) with dry friction with a load of $F = 0.8$ MPa; (b) with friction with lubrication at $F = 15$ MPa. The numbers indicate AMC: 1-AL25 + 5% SiC, 2-D16 + 20% SiC, 3-AK12 + 5% SiC, 4-D16 + 5% SiC.
3.4 Research in some industrial friction units replacing parts from traditional materials with parts from AMC

From the obtained AMCs, some model parts for real friction units were obtained and the results of replacing parts from traditional materials with parts from AMC were analyzed. The results of these studies are shown in Table 3.

The data in Table 3 demonstrate, using the example of some real products in service, the clear advantages of the AMC of the group in question, compared to replaceable materials. Quotas of superiority in terms of durability, service life, weight, and cost have been reached. All presented units have passed bench and field tests. Thus, in practice, it has been shown that parts made from AMC have higher performance characteristics. In all the examples presented, due to the replacement of the material, a significant increase in the service life has been achieved—by 1.4 ... 1.5 times. A threefold reduction in part weight can be achieved. According to the economic factor, the gain is usually 2 to 10 times. These results are in good agreement with the literature data. In particular, works [27], [28] also indicated a decrease in the weight and cost of products made of similar AMCs, and articles [18], [23]-[24], [28]-[32] indicated an increase in the service life of parts from AMCs with similar composition. It should be noted that AMC, thanks to an aluminum-based matrix, is resistant to chemical corrosion, including in seawater. In general, it can be noted that the introduction of AMC is most promising for parts of friction units in the automotive, aerospace, shipbuilding, and oil-producing industries, where, along with high tribotechnical characteristics, weight reduction and good corrosion resistance are required.

4. Conclusions

Many composites are based on aluminum alloys AK12, D16, W124, AL9, AL25, reinforced with silicon carbide SiC particles in an amount from 5 to 20 vol.% obtained by the technology of liquid-phase mechanical mixing. The developed composites in comparison with the initial matrix alloys had 1.1 ... 1.4 times higher values of hardness, with a fairly uniform distribution of their values over the cross-section of the sample.

Under conditions of dry friction on steel according to the “disk-block” loading scheme in the range of rotation speeds of 300-1,500 rpm and loads of 70-300 N, the composites showed better efficiency compared to matrix alloys and the frequently used anti-frictional aluminum alloy AOM 2-1. The dry friction coefficient was 1.5-3 times lower on average, and the volumetric wear rate was 5-9 times lower. An increase in the content of SiC particles in the composite from 5 to 10-20 vol.%. Usually leads to an improvement in tribotechnical characteristics. The composites obtained have shown a sufficiently high operational suitability for work in friction units with steels both in dry friction and in friction with lubrication.

Of the composites studied, the best performance characteristics and consequently, the greatest promising application in the industry were shown by the materials D16 + 20% SiC and AL25 + 10% SiC. Composites W124 + 15% SiC and AK12 + 10% SiC also have a potential for use.

Full-scale tests on replacing parts made of traditional anti-friction materials with parts made from obtained composites in real friction units have shown that such a replacement makes it possible to improve the performance, increase the service life, and reduce the cost of products by 2-10 times.

References


2016.


