



## Research Article

# Unveiling Eco-Friendly Energy Storage Using (Ge/Sn/Pb)-Doped SiC Nanocomposites Through Hydrogen Adsorption: Materials Modeling by DFT Study

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**Abstract:** As applied materials for quantum nanotechnology, silicon carbide nanocage (Si-C<sub>NC</sub>), germanium carbide nanocage (Ge-C<sub>NC</sub>), tin carbide nanocage (Sn-C<sub>NC</sub>) or lead carbide nanocage (Pb-C<sub>NC</sub>) have attracted considerable attention in materials science. A comprehensive investigation on hydrogen grabbing by main group carbides of Si, Ge, Sn or Pb was carried out including using density functional theory (DFT) computations at the Coulomb-Attenuating method with the hybrid functional Becke 3-parameter Lee-Yang-Parr, Electron Paramagnetic Resonance (CAM-B3LYP)/EPR-III), 6-311+G (d, p), and Los Alamos National Laboratory 2 double  $\zeta$  (LANL2DZ) level of theory. The data represents that if silicon elements are replaced by germanium, tin or lead, the H-grabbing energy will be ameliorated. Electromagnetic and thermodynamic properties of Si-C<sub>NC</sub>, Ge-C<sub>NC</sub>, Sn-C<sub>NC</sub>, Pb-C<sub>NC</sub> and H@Si-C<sub>NC</sub>, H@Ge-C<sub>NC</sub>, H@Sn-C<sub>NC</sub>, H@Pb-C<sub>NC</sub> clusters were analyzed. The hypothesis of the adsorption phenomenon was confirmed by density distributions of the density of states (DOS) and partial density of states (PDOS) and charge distribution. The fluctuation in charge density values demonstrates that the electronic densities were mainly located in the boundary of adsorbate/adsorbent atoms during the adsorption status. All in all, the consequences display that the Si-C<sub>NC</sub>, Ge-C<sub>NC</sub>, Sn-C<sub>NC</sub> or Pb-C<sub>NC</sub> might be appropriate candidate nanocones for hydrogen storage. This study proves that the application of silicon carbide, germanium carbide, tin carbide or lead carbide nanocages in supercapacitor devices has great potential in the field of energy storage, providing a reference for the further development of novel semiconductors in the field of energy storage and optoelectronic devices.

**Keywords:** hydrogen storage, DFT computation, carbide nanocage

## 1. Introduction

Designing advanced electrode materials that can be reliably cycled at high temperatures and used for assembling advanced energy storage devices remains a major challenge. As a representative of novel wide bandgap semiconductors, silicon carbide, germanium carbide, tin carbide or lead carbide nanocages have broad prospects in energy storage due to their excellent characteristics of stable chemical properties.

A type of clean fuel is hydrogen that might be employed to accumulate, carry, and spread energy produced by other sources and it generates water when applied in a fuel cell.<sup>1</sup> A fuel cell applies reverse electrolysis to convert an oxidizing agent and hydrogen to power an electric motor.<sup>2-4</sup>

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CNTs owing to their lightness, tube construction, vast plane and high reactivity between C and H atoms can be proposed as an estimated material for H-grabbing.<sup>5-8</sup> Biofuel cells have been in the spotlight for the past century because of their potential and promise as a unique platform for sustainable energy harvesting from the human body and the environment. Because biofuel cells are typically developed in a small platform serving as a primary battery with limited fuel or as a rechargeable battery with repeated refueling, they have been interchangeably named biobatteries.<sup>9-12</sup>

It was investigated that H-storing on the C-nano compound indicates molecular hydrogen dissociation.<sup>13-16</sup> The structure of transition metal-carbon exhibits a charge distribution among boundary atoms and the cationic state of transition metals can be discussed.<sup>17-21</sup> Thus, the electronic charge can be produced through gas molecules adsorption on the surfaces of ionic transition metal.<sup>22-24</sup> Transition metals as dopants might make a whole Hamiltonian perturbation towards alterations in electronic structures, which convert it a substantial usage in magnetic electronic instruments.<sup>25-29</sup> Recently, Si-, Ge- or Sn-carbide nanostructures have been suggested as engaged H-grabbing compounds.<sup>30-32</sup> Since the polarizability of silicon is more than carbon, it is supposed that Si-C/Si nanosheet might attach to compositions more strongly in comparison to the net carbon nano-surfaces.<sup>33-35</sup>

Therefore, during this work we are going to apply computational methods towards estimating the efficiency of Si-C\_NC, Ge-C\_NC, Sn-C\_NC and Pb-C\_NC clusters for H-grabbing. The calculation has measured the magnetic and electronic structures of H@Si-C\_NC, H@Ge-C\_NC, H@Sn-C\_NC and H@Pb-C\_NC clusters by DFT study,<sup>36</sup> generalized gradient approximation (GGA) potential and Perdew-Burke-Ernzerhof (PBE) functional.<sup>37,38</sup> Moreover, it must be remarked that metal hydrides are still in their early steps of development, and further investigation is needed to evaluate their commercial viability.<sup>39-44</sup>

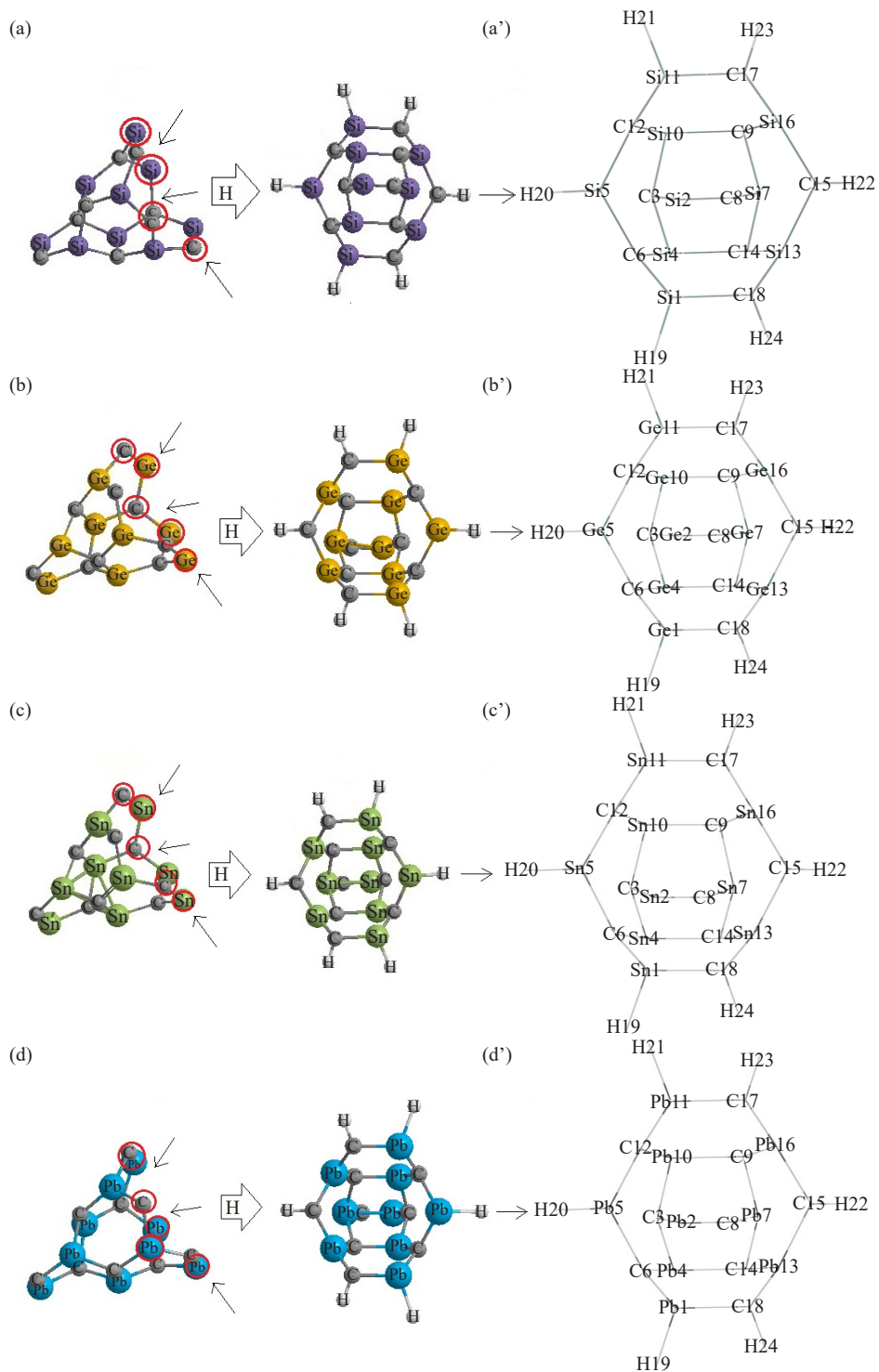
This work wants to construct Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC complexes through hydrogen adsorption calculations. The article intends to illustrate the technical part of computation, and investigate the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC complexes. Then, hydrogen storage is reported through hydrogen adsorption calculations and employing molecular electrostatic potential energy and molecular orbitals computations.

## 2. Theoretical conception, materials, and methods

In Figure 1, nano-carbides of silicon, germanium, tin or lead have been designed by disclination angle of 180°, cone height of 20 Å and MWNC (1, 5).

The H-grabbing on the surface of Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC indicated the chemisorption between H-atoms and Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC using the GaussView 6.06.16<sup>45</sup> and Gaussian 16, Revision C.01<sup>46</sup> and DFT computations at the CAM-B3LYP/EPR-III, 6-311+G (d, p), LANL2DZ level of theory.<sup>47,48</sup>

The charge transfer between H-atoms and Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC is computed by Bader charge parameter.<sup>49-51</sup> The alterations of charge density have demonstrated that Si-C\_NC shows the Bader charge of -0.667 coulomb, before H-adsorption, and -0.674 coulomb after H-adsorption, respectively. Ge-C represents the Bader charge of -0.631 coulomb before H-adsorption, and -0.652 coulomb after H-adsorption, respectively. Sn-C indicates the Bader charge of -0.829 coulomb before H-adsorption, and -0.858 coulomb after H-adsorption, respectively. Pb-C has the Bader charge of -0.913 coulomb before H-adsorption, and -0.923 coulomb after H-adsorption, respectively. Thus, the alterations of charge density for H-grabbing on Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC are  $\Delta Q_{H@Si-C\_NC} = -0.007e > \Delta Q_{H@Sn-C\_NC} = -0.029e > \Delta Q_{H@Ge-C\_NC} = -0.021e > \Delta Q_{H@Pb-C\_NC} = -0.01e$ . Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC have played the role of electron acceptors while H-atoms as the stronger electron donors have appeared through adsorption on these nano-carbides. In fact, diffusion of the H atoms from the most stable adsorption site to the nearest neighbouring site was studied in order to obtain diffusion barrier and diffusion values.

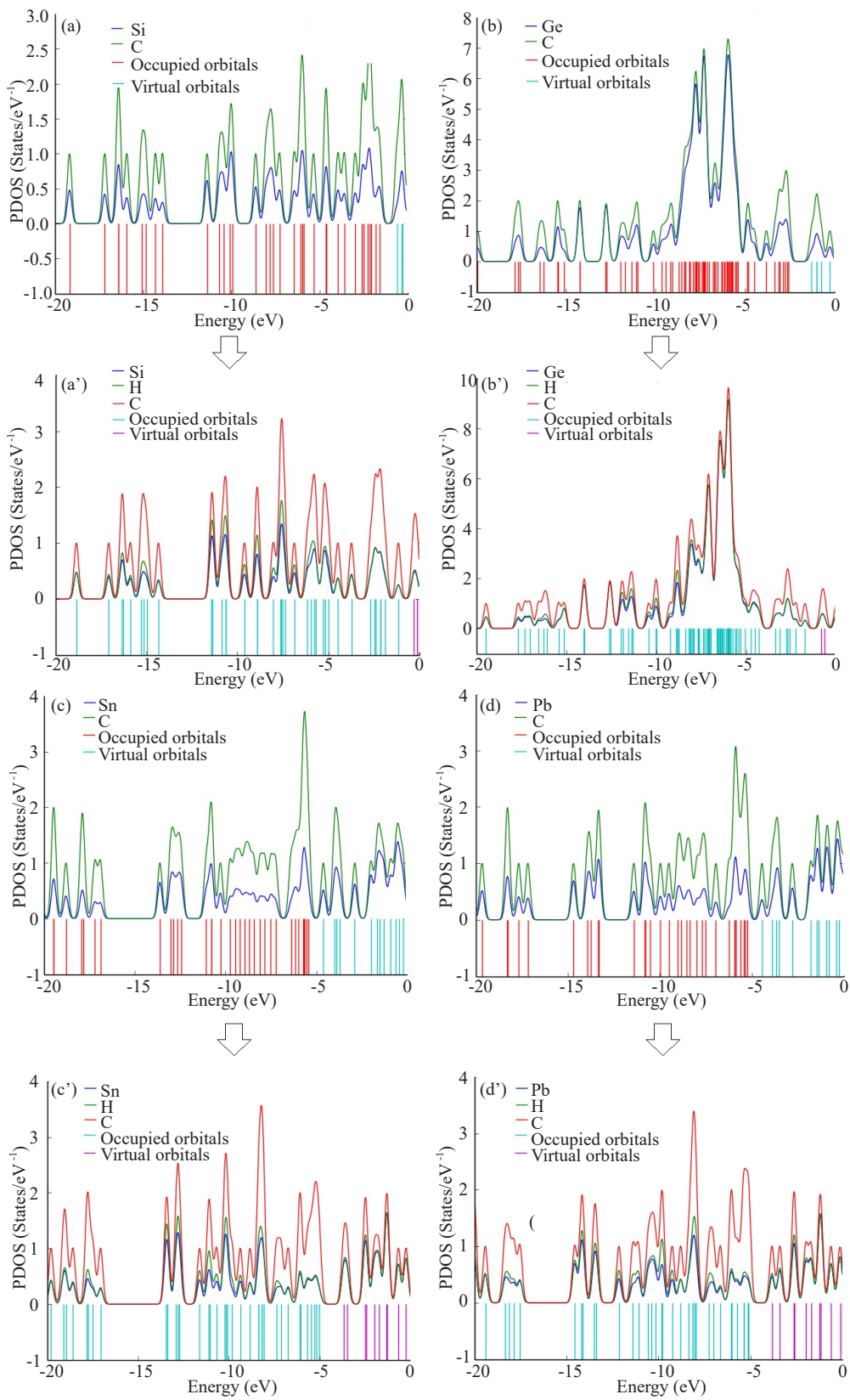


**Figure 1.** H-grabbing by (a) Si-C\_NC, (b) Ge-C\_NC, (c) Sn-C\_NC, (d) Pb-C\_NC towards formation of (a') H@Si-C\_NC, (b') H@Ge-C\_NC, (c') H@Sn-C\_NC, and (d') H@Pb-C\_NC

### 3. Results and discussions

#### 3.1 Electronic characterization of nano-carbides

It is clear from Figure 2 that after H-grabbing by Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC, there is a significant



**Figure 2.** PDOS adsorption of (a) Si-C\_NC, (a') H@Si-C\_NC, (b) Ge-C\_NC, (b') H@Ge-C\_NC, (c) Sn-C\_NC, (c') H@Sn-C\_NC, (d) Pb-C\_NC and (d') H@Pb-C\_NC complexes

contribution of Ge, Sn, Pb d-orbital in the unoccupied level. Density of states (DOS) resulted that Si, Ge, Sn, Pb stay in the ionic form towards admitting more electrons from other particles (Figure 2a-d, a'-d').

Figure 2(a) and (a') shows that the Si-C\_NC and H@Si-C\_NC states, respectively, have more contribution at the conduction band around -5 to -15 eV, while contribution of C-states is wide, and Si- and H-states possess less contributions. The nanostructures of Ge-C\_NC (Figure 2b), H@Ge-C\_NC (Figure 2b'), Sn-C\_NC (Figure 2c), H@Sn-C\_NC (Figure 2c'), Pb-C\_NC (Figure 2d) and H@Pb-C\_NC (Figure 2d'), respectively, possess high contribution at the middle of the conduction band between -5 to -10 eV, while contribution of carbon states are wide with high distribution around -7 eV, and Ge-, Sn- or Pb-states have less contributions in comparison to C-states. The projected density of states shows that the H atoms with 1s orbital electrons will hybridize with C/Si/Ge/Sn/Pb atoms with the 2p/3p/3d/4d/5d for orbital electrons of the nearby surface. The climbing image nudged elastic band study shows that the surface free H atoms are very easy to diffuse between the surface short-bridge sites and the subsurface short-bridge sites but the diffusion between the short-bridge site and the triangular center site is extremely difficult.

### 3.2 NQR analysis

NQR parameters of gas molecules of H-grabbing on Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC based on cartesian harmonics are:<sup>52</sup>

$$V(r) = V(0) + \left[ \left( \frac{\partial V}{\partial x_i} \right) \Big|_0 \cdot x_i \right] + \frac{1}{2} \left[ \left( \frac{\partial^2 V}{\partial x_i \partial x_j} \right) \Big|_0 \cdot x_i x_j \right] + \dots \quad (1)$$

$$U = -\frac{1}{2} \int_{\mathcal{D}} d^3r \rho_r \left[ \left( \frac{\partial^2 V}{\partial x_i^2} \right) \Big|_0 \cdot x_i^2 \right] = -\frac{1}{2} \int_{\mathcal{D}} d^3r \rho_r \left[ \left( \frac{\partial E_i}{\partial x_i} \right) \Big|_0 \cdot x_i^2 \right] = -\frac{1}{2} \left( \frac{\partial E_i}{\partial x_i} \right) \Big|_0 \cdot \int_{\mathcal{D}} d^3r [\rho(r) \cdot x_i^2] \quad (2)$$

$$\chi = e^2 Q q_{zz} / h \quad (3)$$

$$\eta = q_{xx} - q_{yy} / q_{zz} \quad (4)$$

where  $\chi$ , quadrupole coupling constant;  $\eta$ , asymmetry factor of the electric field gradient tensor;  $q_{ii}$ , ingredients of the electric field gradient tensor;  $e$ , the proton charge;  $h$ , the Planck's constant and  $Q$ , the nuclear quadrupole moment are the NQR parameters.<sup>52</sup>

**Table 1.** The parameters of electric potential ( $E_p$ ) and Bader charge ( $Q$ ) of H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC using “CAM-B3LYP/EPR-III, 6-311+G (d, p), LANL2DZ” computations

Si-C_NC			H@Si-C_NC			Ge-C_NC			H@Ge-C_NC		
Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$
Si(1)	0.3110	-48.583	Si(1)	0.4335	-48.5485	Ge(1)	0.2338	-153.986	Ge(1)	0.3472	-153.99
Si(2)	0.5853	-48.6074	Si(2)	0.5886	-48.6244	Ge(2)	0.4877	-153.629	Ge(2)	0.4919	-153.637
C(3)	-0.4429	-14.6956	C(3)	-0.4439	-14.6901	C(3)	-0.3721	-14.634	C(3)	-0.3724	-14.6381
Si(4)	0.4601	-48.4976	Si(4)	0.4694	-48.5212	Ge(4)	0.4184	-153.917	Ge(4)	0.3929	-153.941

Table 1. (cont.)

Si-C_NC			H@Si-C_NC			Ge-C_NC			H@Ge-C_NC		
Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$
Si(5)	0.3550	-48.5651	Si(5)	0.4596	-48.5393	Ge(5)	0.3336	-154.015	Ge(5)	0.3576	-153.98
C(6)	-0.4512	-14.7206	C(6)	-0.4765	-14.7486	C(6)	-0.4001	-14.6755	C(6)	-0.4081	-14.7029
Si(7)	0.6123	-48.4375	Si(7)	0.6281	-48.448	Ge(7)	0.6092	-153.911	Ge(7)	0.6049	-153.922
C(8)	-0.6666	-14.7584	C(8)	-0.6739	-14.7642	C(8)	-0.6311	-14.6862	C(8)	-0.6313	-14.6887
C(9)	-0.4844	-14.7239	C(9)	-0.5115	-14.7378	C(9)	-0.4373	-14.6715	C(9)	-0.4596	-14.6975
Si(10)	0.4601	-48.4976	Si(10)	0.4694	-48.5212	Ge(10)	0.4184	-153.917	G(10)	0.3929	-153.941
Si(11)	0.3110	-48.583	Si(11)	0.4335	-48.5485	Ge(11)	0.2338	-153.986	Ge(11)	0.3472	-153.99
C(12)	-0.4512	-14.7206	C(12)	-0.4765	-14.7486	C(12)	-0.4001	-14.6755	C(12)	-0.4081	-14.7029
Si(13)	0.4501	-48.5069	Si(13)	0.5577	-48.5043	Ge(13)	0.3916	-153.938	Ge(13)	0.4952	-153.976
C(14)	-0.4844	-14.7239	C(14)	-0.5115	-14.7378	C(14)	-0.4373	-14.6715	C(14)	-0.4596	-14.6975
C(15)	-0.3158	-14.7558	C(15)	-0.4042	-14.7189	C(15)	-0.2627	-14.7083	C(15)	-0.3785	-14.6938
Si(16)	0.4501	-48.5069	Si(16)	0.5577	-48.5043	Ge(16)	0.3916	-153.938	Ge(16)	0.4952	-153.976
C(17)	-0.3491	-14.7754	C(17)	-0.4122	-14.7179	C(17)	-0.2887	-14.7187	C(17)	-0.3626	-14.677
C(18)	-0.3491	-14.7754	C(18)	-0.4122	-14.7179	C(18)	-0.2887	-14.7187	C(18)	-0.3626	-14.677
			H(19)	-0.1245	-1.2466				H(19)	-0.0808	-1.2055
			H(20)	-0.1254	-1.2421				H(20)	-0.0793	-1.1995
			H(21)	-0.1245	-1.2466				H(21)	-0.0808	-1.2055
			H(22)	0.0330	-1.1770				H(22)	0.0505	-1.1604
			H(23)	0.0332	-1.1951				H(23)	0.0540	-1.1689
			H(24)	0.0332	-1.1951				H(24)	0.0540	-1.1689
Sn-C_NC			H@Sn-C_NC			Pb-C_NC			H@Pb-C_NC		
Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$
Sn(1)	0.3787	-284.1396	Sn(1)	0.5724	-284.114	Pb(1)	0.4641	-1.6181	Pb1	0.5181	-1.6193
Sn(2)	0.6382	-283.7272	Sn(2)	0.6699	-283.74	Pb(2)	0.1438	-1.5339	Pb2	0.1552	-1.5519
C(3)	-0.6662	-14.73518	C(3)	-0.6531	-14.7512	C(3)	-0.9130	-14.787	C3	-0.9228	-14.8081
Sn(4)	0.6680	-284.0589	Sn(4)	0.6423	-284.081	Pb(4)	0.8615	-1.5799	Pb4	0.8515	-1.5982
Sn(5)	0.4801	-284.1451	Sn(5)	0.6068	-284.111	Pb(5)	0.6254	-1.6217	Pb5	0.6371	-1.6140
C(6)	-0.6210	-14.7433	C(6)	-0.6554	-14.7892	C(6)	-0.8941	-14.7884	C6	-0.9032	-14.8208
Sn(7)	0.8287	-284.0484	Sn(7)	0.8584	-284.056	Pb(7)	0.7935	-1.5598	Pb7	0.7357	-1.5792
C(8)	-0.7438	-14.6500	C(8)	-0.7563	-14.6564	C(8)	0.0335	-14.76	C8	0.0352	-14.778
C(9)	-0.6772	-14.7617	C(9)	-0.6950	-14.7774	C(9)	-0.7484	-14.8077	C9	-0.7587	-14.8172

Table 1. (cont.)

Sn-C_NC			H@Sn-C_NC			Pb-C_NC			H@Pb-C_NC		
Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$	Atom	$Q$	$E_p$
Sn(10)	0.6680	-284.0589	Sn(10)	0.6423	-284.081	Pb(10)	0.8615	-1.5799	Pb10	0.8530	-1.5981
Sn(11)	0.3787	-284.1396	Sn(11)	0.5724	-284.114	Pb(11)	0.4641	-1.6181	Pb11	0.5203	-1.6191
C(12)	-0.6210	-14.7433	C(12)	-0.6554	-14.7892	C(12)	-0.8941	-14.7884	C12	-0.9039	-14.8207
Sn(13)	0.6552	-284.0813	Sn(13)	0.7660	-284.101	Pb(13)	0.6859	-1.5973	Pb13	0.8239	-1.6032
C(14)	-0.6772	-14.7617	C(14)	-0.6950	-14.7774	C(14)	-0.7484	-14.8077	C14	-0.7586	-14.8172
C(15)	-0.4494	-14.7758	C(15)	-0.5776	-14.7784	C(15)	-0.3692	-14.7679	C15	-0.6837	-14.8006
Sn(16)	0.6552	-284.0813	Sn(16)	0.7660	-284.101	Pb(16)	0.6859	-1.5973	Pb16	0.8243	-1.6030
C(17)	-0.4476	-14.7665	C(17)	-0.5268	-14.7389	C(17)	-0.526	-14.7799	C17	-0.7146	-14.7992
C(18)	-0.4476	-14.7665	C(18)	-0.5268	-14.7389	C(18)	-0.526	-14.7799	C18	-0.7128	-14.7992
			H(19)	-0.1566	-1.23558				H19	-0.0690	-1.0892
			H(20)	-0.1507	-1.22487				H20	-0.0674	-1.0899
			H(21)	-0.1566	-1.23558				H21	-0.0743	-1.0893
			H(22)	0.0323	-1.20447				H22	0.2066	-1.0660
			H(23)	0.0383	-1.20075				H23	0.2043	-1.0864
			H(24)	0.0383	-1.20075				H24	0.2036	-1.0864

Table 1 has shown the nuclear quadrupole resonance for H-grabbing on the nano-carbides of Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC and formation of H@Si-C\_NC, H@Ge-C\_NC, H@Sn-C\_NC, H@Pb-C\_NC clusters.

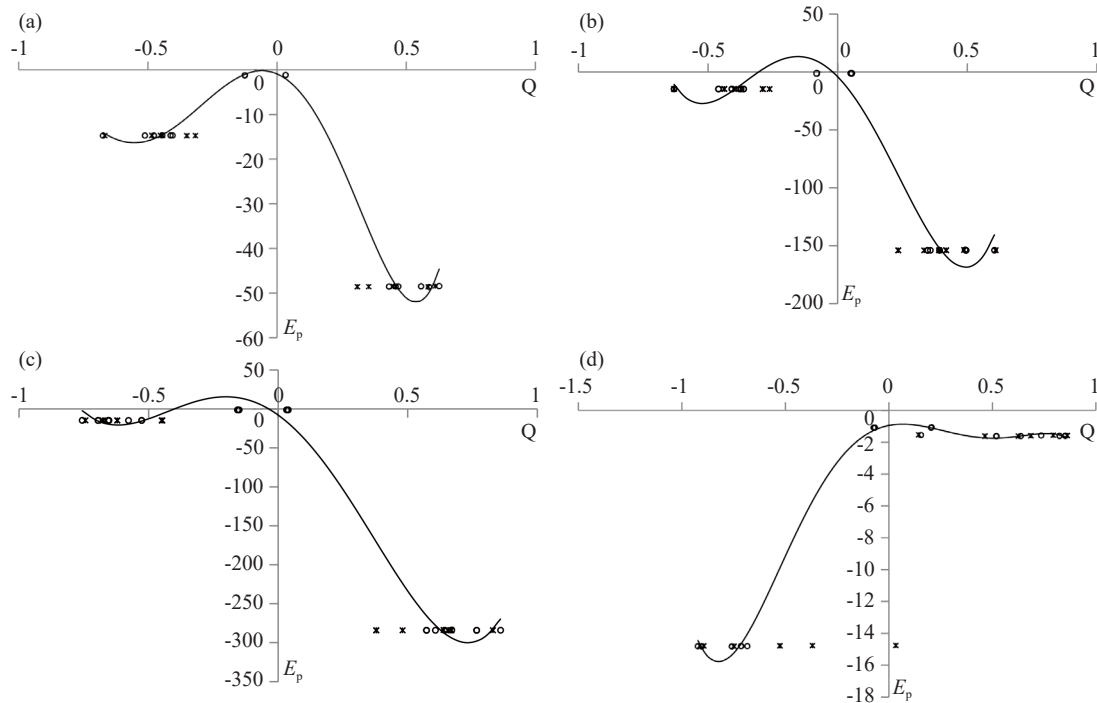


Figure 3. “NQR” parameters for (a) H@Si-C\_NC, (b) H@Ge-C\_NC, (c) H@Sn-C\_NC, and (d) H@Pb-C\_NC clusters



Furthermore, the electric potential of nuclear quadrupole resonance method versus Bader charge for H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC by “CAM-B3LYP/EPR-III, 6-311+G (d, p), LANL2DZ” theoretical level has been shown (Figure 3a-d). As the number of calculation studies related to metals and metalloids has been enhancing, the “LANL2DZ” basis set has become one of the most common basis sets for metal and metalloid complexes among others. “EPR-III” is a triple zeta basis set consisting of diffuse functions, double d-polarizations and a single set of f-polarization functions. In this case, the s-part is developed to better illustrate the nuclear zone.

It has been observed the impact of the binding between hydrogen atoms with C, Si, Ge, Sn and Pb in the H@Si-C\_NC, H@Ge-C\_NC, H@Sn-C\_NC, H@Pb-C\_NC during H-grabbing through “NQR” analysis (Figure 3a-d). The plots of H@Si-C\_NC (Figure 3a), H@Ge-C\_NC (Figure 3b), H@Sn-C\_NC (Figure 3c), and H@Pb-C (Figure 3d), respectively, have illustrated the impact of H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC and Pb-C\_NC clusters. It is suitable for the study of measured nanoclusters containing quadrupole nuclei where the characteristic NQR frequencies represent the fingerprints of these compounds. In several cases, <sup>14</sup>N NQR can distinguish between the crystalline phases of hydrogenated surfaces. So, it can be stated that <sup>14</sup>N NQR is a powerful contactless tool for detecting the appearance of H-grabbing with a possibility to distinguish clearly and quantitatively among different nanoclusters.

### 3.3 Magnetism of H-adsorption on Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC

“NMR” parameters for certain atoms in the active site of H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC through the linkage between H-atoms and nano-carbides towards forming H@Si-C\_NC, H@Ge-C\_NC, H@Sn-C\_NC, H@Pb-C\_NC hydrides have been measured in Table 2. The density functional theory (DFT) calculations have been executed using the Gaussian 16 package.<sup>46</sup>

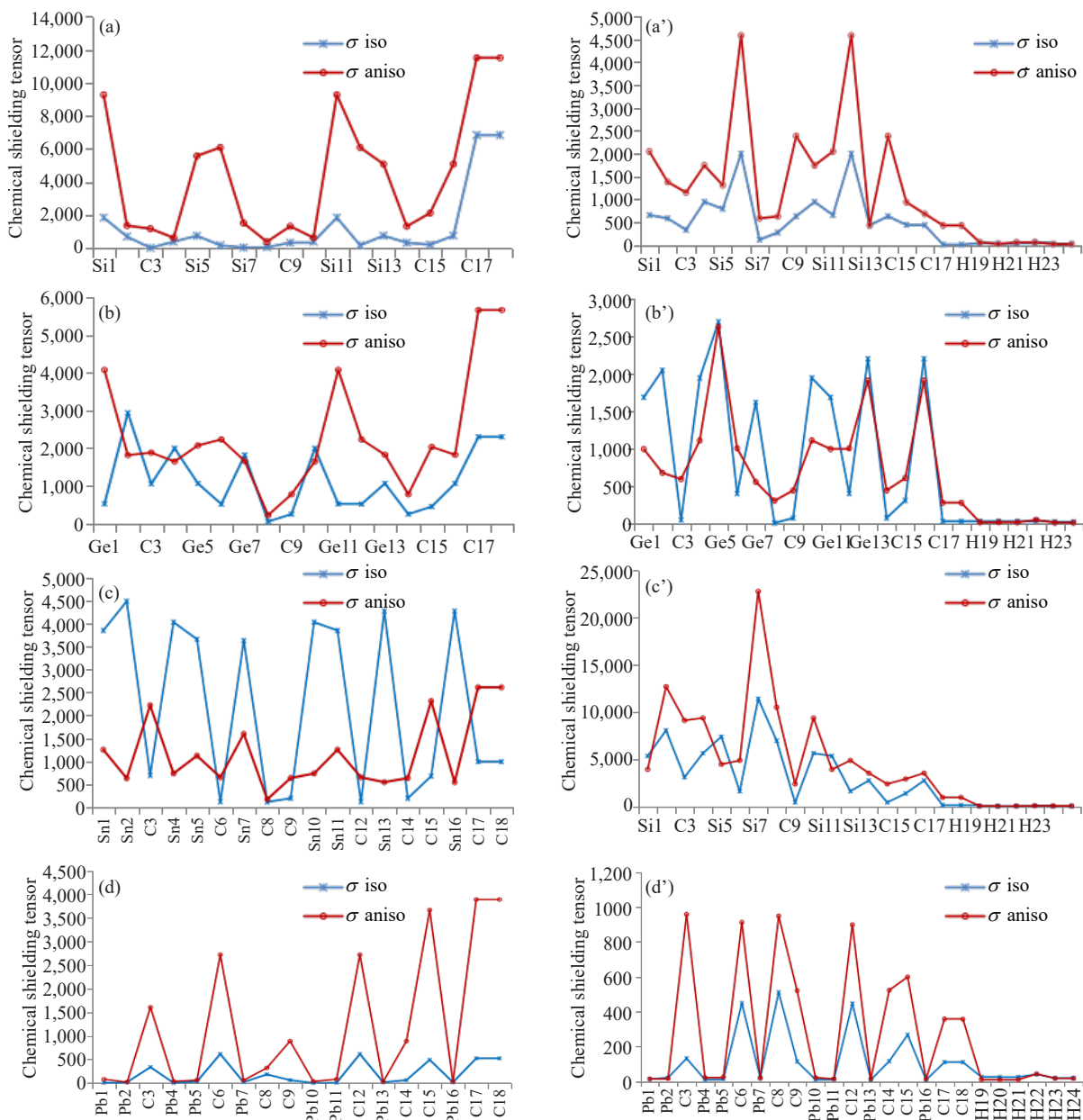
**Table 2.** “NMR” parameters of H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC clusters

Si-C_NC			H@Si-C_NC			Ge-C_NC			H@Ge-C_NC		
Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$
Si(1)	1,852.8960	9,277.8581	Si(1)	652.0996	2,043.4448	Ge(1)	524.2652	4,076.8426	Ge(1)	1,685.9528	995.5874
Si(2)	707.1922	1,372.5500	Si(2)	578.8802	1,376.1814	Ge(2)	2,938.4569	1,816.3038	Ge(2)	2,046.7422	677.2387
C(3)	26.3336	1,184.7146	C(3)	328.7614	1,145.6732	C(3)	1,055.1306	1,882.3207	C(3)	45.4161	593.3670
Si(4)	396.8693	636.0558	Si(4)	939.4444	1,737.4509	Ge(4)	1,995.8705	1,646.8872	Ge(4)	1,943.3848	1,109.6931
Si(5)	751.2993	5,584.1673	Si(5)	787.0425	1,304.9949	Ge(5)	1,063.4297	2,074.3377	Ge(5)	2,694.1662	2,627.1478
C(6)	180.1857	6,093.0613	C(6)	1,995.0521	4,581.1275	C(6)	517.8104	2,231.1687	C(6)	398.1112	1,004.5925
Si(7)	44.5656	1,523.6745	Si(7)	112.7493	579.3102	Ge(7)	1,817.7074	1,660.1330	Ge(7)	1,618.7559	555.2137
C(8)	62.5710	392.1047	C(8)	269.5126	621.8678	C(8)	54.3886	222.9889	C(8)	6.1355	304.2995
C(9)	334.2540	1,328.9150	C(9)	624.3653	2,381.2166	C(9)	250.6261	776.3827	C(9)	73.4150	442.0441
Si(10)	396.8693	636.0558	Si(10)	939.4444	1,737.4509	Ge(10)	1,995.8705	1,646.8872	G(10)	1,943.3848	1,109.6931
Si(11)	1,852.8960	9,277.8581	Si(11)	652.0996	2,043.4448	Ge(11)	524.2652	4,076.8426	Ge(11)	1,685.9528	995.5874
C(12)	180.1857	6,093.0613	C(12)	1,995.0521	4,581.1275	C(12)	517.8104	2,231.1687	C(12)	398.1112	1,004.5925
Si(13)	758.7227	5,091.8836	Si(13)	428.6833	428.6833	Ge(13)	1,064.2837	1,824.8636	Ge(13)	2,199.6950	1,909.5554



Table 2. (cont.)

Si-C_NC			H@Si-C_NC			Ge-C_NC			H@Ge-C_NC		
Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$
C(14)	334.2540	1,328.9150	C(14)	624.3653	2,381.2166	C(14)	250.6261	776.3827	C(14)	73.4150	442.0441
C(15)	215.4441	2,129.1385	C(15)	434.8456	933.7457	C(15)	451.4650	2,033.6451	C(15)	308.3324	606.6501
Si(16)	758.7227	5,091.8836	Si(16)	428.6833	678.2943	Ge(16)	1,064.2837	1,824.8636	Ge(16)	2,199.6950	1,909.5554
C(17)	6,842.2257	11,517.1989	C(17)	3.3059	424.8038	C(17)	2,301.7865	5,664.7386	C(17)	29.3094	277.5181
C(18)	6,842.2257	11,517.1989	C(18)	3.3059	424.8038	C(18)	2,301.7865	5,664.7386	C(18)	29.3094	277.5181
			H(19)	35.7119	53.7922				H(19)	28.7663	15.3241
			H(20)	22.7277	19.1040				H(20)	28.5430	18.0659
			H(21)	35.7119	53.7922				H(21)	28.7663	15.3241
			H(22)	45.5793	57.7040				H(22)	41.0031	48.3955
			H(23)	18.3406	13.9135				H(23)	21.0924	11.7482
			H(24)	18.3406	13.9135				H(24)	21.0924	11.7482
Sn-C_NC			H@Sn-C_NC			Pb-C_NC			H@Pb-C_NC		
Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$	Atom	$\sigma_{iso}$	$\sigma_{aniso}$
Sn(1)	3,852.4768	1,253.8614	Sn(1)	5,331.1122	3,908.8109	Pb(1)	8.3138	81.0012	Pb(1)	12.3612	13.2040
Sn(2)	4,488.4542	630.7182	Sn(2)	8,048.7803	12,679.4768	Pb(2)	21.0773	19.9446	Pb(2)	20.9858	15.8398
C(3)	692.2861	2,220.5600	C(3)	3,086.7050	9,101.5478	C(3)	337.2144	1,607.1525	C(3)	131.0567	958.8770
Sn(4)	4,033.8531	734.5909	Sn(4)	5,625.1352	9,355.5742	Pb(4)	1.8733	35.7403	Pb(4)	9.3660	21.0628
Sn(5)	3,660.3970	1,122.0832	Sn(5)	7,373.9159	4,446.7111	Pb(5)	33.6067	62.8027	Pb(5)	12.5136	24.9133
C(6)	117.5385	651.1418	C(6)	1,601.6375	4,860.7895	C(6)	615.7565	2,721.2686	C(6)	448.3251	912.9272
Sn(7)	3,631.0020	1,598.1063	Sn(7)	11,387.7893	22,754.5333	Pb(7)	19.1565	58.8387	Pb(7)	20.4001	17.7409
C(8)	114.3168	171.0993	C(8)	6,947.4689	10,490.2518	C(8)	184.8183	320.2073	C(8)	510.5573	949.4451
C(9)	192.3590	634.9115	C(9)	419.8777	2,370.5045	C(9)	61.9853	889.5642	C(9)	113.4349	520.4584
Sn(10)	4,033.8531	734.5909	Sn(10)	5,625.1352	9,355.5742	Pb(10)	1.8733	35.7403	Pb(10)	9.2549	21.0870
Sn(11)	3,852.4768	1,253.8614	Sn(11)	5,331.1122	3,908.8109	Pb(11)	8.3138	81.0012	Pb(11)	12.2325	13.1823
C(12)	117.5385	651.1418	C(12)	1,601.6375	4,860.7895	C(12)	615.7565	2,721.2686	C(12)	446.5082	899.4323
Sn(13)	4,274.5080	544.9696	Sn(13)	2,735.2331	3,523.3606	Pb(13)	17.5248	27.0793	Pb(13)	8.0973	13.0066
C(14)	192.3590	634.9115	C(14)	419.8777	2,370.5045	C(14)	61.9853	889.5642	C(14)	115.2409	523.1188
C(15)	677.3063	2,314.0820	C(15)	1,362.9700	2,901.5605	C(15)	492.2542	3,669.8947	C(15)	266.7875	599.4562
Sn(16)	4,274.5080	544.9696	Sn(16)	2,735.2331	3,523.3606	Pb(16)	17.5248	27.0793	Pb(16)	8.0859	12.8627
C(17)	992.3714	2,612.5933	C(17)	125.3188	949.5305	C(17)	523.3055	3,890.6558	C(17)	109.6002	357.9518
C(18)	992.3714	2,612.5933	C(18)	125.3188	949.5305	C(18)	523.3055	3,890.6558	C(18)	109.5379	356.4523
			H(19)	29.6065	44.0962				H(19)	24.8456	9.3054
			H(20)	24.6953	11.3083				H(20)	23.6407	8.6042
			H(21)	29.6065	44.0962				H(21)	24.5059	8.9632
			H(22)	60.6118	70.1035				H(22)	39.9395	42.2828
			H(23)	0.8243	61.1416				H(23)	18.3117	16.5055
			H(24)	0.8243	61.1416				H(24)	18.3067	16.3958



**Figure 4.** Isotropic ( $\sigma_{\text{iso}}$ ) and anisotropy ( $\sigma_{\text{aniso}}$ ) tensors derived from NMR spectrums for (a) Si-C\_NC, (a') H@Si-C\_NC, (b) Ge-C\_NC, (b') H@Ge-C\_NC, (c) Sn-C\_NC, (c') H@Sn-C\_NC, (d) Pb-C\_NC and (d') H@Pb-C\_NC clusters

“NMR” graphs in Figure 4 (a-d, a'-d') have indicated the similar chemical shielding attitude of isotropic and anisotropy factors of Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC and H@Si-C\_NC, H@Ge-C\_NC, H@Sn-C\_NC, H@Pb-C\_NC clusters with several sharp peaks related to C, Si, Ge, Sn, Pb “adsorbent” and H-atoms “adsorbate” in the active zones.<sup>53,54</sup>

Adsorption “NMR” graphs of Si1, Si5, Si11, C6, C17, C18 for SiC (Figure 4a) and C6, C12 for H@SiC (Figure 4a'); Ge1, Ge11, C17, C18 for GeC (Figure 4b) and Ge2, Ge5, Ge7, Ge10, Ge13, Ge16 for H@GeC (Figure 4b'); Sn2, Sn4, Sn5, Sn7, Sn10, Sn11, Sn13, Sn16 for SnC (Figure 4c) and Sn2 for H@SnC (Figure 4c'); C3, C6, C12, C15, C17, C18 for PbC (Figure 4d) and C3, C6, C8, C12 for H@PbC (Figure 4d'), respectively have been indicated.

### 3.4 Vibrations of H-adsorption onto Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC

Concerning adsorption process, the thermodynamic parameters were measured for H-grabbing on the nanocarbitides of Si-C\_NC, Ge-C\_NC, Sn-C\_NC, Pb-C\_NC as the gas sensors which might be suitable for H-storing (Table 3).

**Table 3.** The thermodynamic properties of Si-C\_NC, H@Si-C\_NC, Ge-C\_NC, H@Ge-C\_NC, Sn-C\_NC, H@Sn-C\_NC, Pb-C\_NC and H@Pb-C\_NC complexes

Nanocone and H-adsorption	$\Delta E^\circ \times 10^{-4}$ (kcal/mol)	$\Delta H^\circ \times 10^{-4}$ (kcal/mol)	$\Delta G^\circ \times 10^{-4}$ (kcal/mol)	$S^\circ$ (Cal/K·mol)	Dipole moment (Debye)	Convergence	k-point grid
Si-C_NC	-182.7474	-182.7474	-182.7514	134.576	4.1234	0.7415D-08	2.0062
H@Si-C_NC	-182.9776	-182.9775	-182.9815	132.507	2.3194	0.7508D-08	2.0154
Ge-C_NC	-1,180.6421	-1,180.6421	-1,180.6465	148.277	3.3993	0.5017D-08	2.0007
H@Ge-C_NC	-1,180.8751	-1,180.8750	-1,180.8798	161.024	1.4991	0.5591D-08	2.0021
Sn-C_NC	-3,390.0971	-3,390.0970	-3,390.1014	147.503	3.0569	0.7533D-08	2.0003
H@Sn-C_NC	-3,390.3269	-3,390.3269	-3,390.3316	157.153	2.1738	0.7927D-08	2.0009
Pb-C_NC	-23.3096	-23.3095	-23.3132	123.440	8.6603	0.8073D-08	1.9967
H@Pb-C_NC	-23.5365	-23.5365	-23.5405	136.838	3.1099	0.8236D-08	2.0001

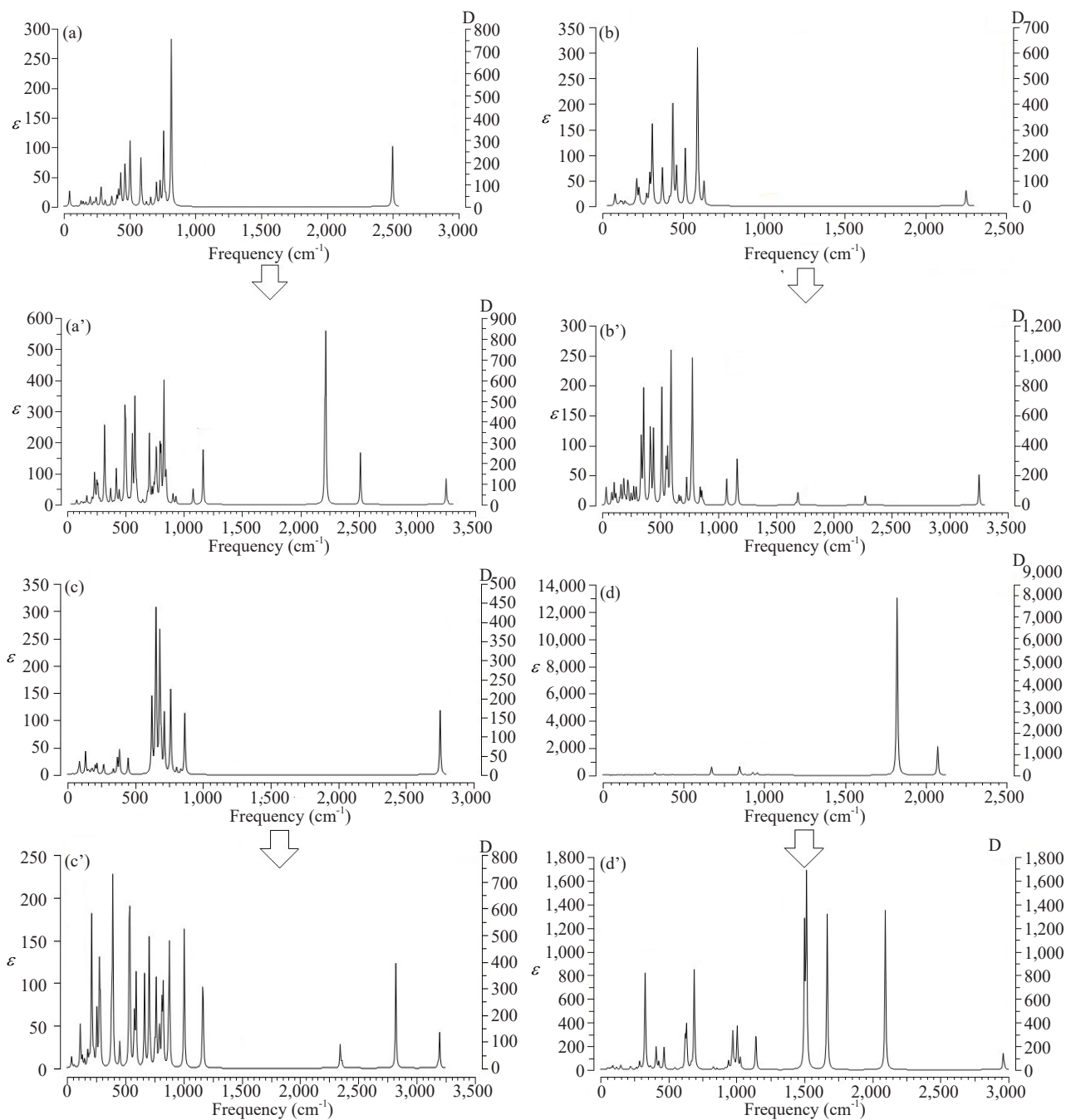
Furthermore, the “IR” spectra for H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC clusters have been seen in Figure 5 (a-d, a'-d'). The curve of Figure 5 (a) and (a') has been seen in the frequency range between 500-2,500  $\text{cm}^{-1}$  for the complex of Si-C\_NC and H@Si-C\_NC with pointed peaks near 815.93  $\text{cm}^{-1}$  and 227.03  $\text{cm}^{-1}$ , respectively. The graph of Figure 5 (b) and (b') has been found in the frequency range between 500-1,000  $\text{cm}^{-1}$  for the complex of Ge-C\_NC and H@Ge-C\_NC with pointed peaks near 588.37  $\text{cm}^{-1}$  and 777.77  $\text{cm}^{-1}$ , respectively. The curve of Figure 5 (c) and (c') has been indicated in the frequency range between 250-1,250  $\text{cm}^{-1}$  for the complex of Sn-C\_NC and H@Sn-C\_NC with pointed peaks near 761.38  $\text{cm}^{-1}$  and 393.08  $\text{cm}^{-1}$ , respectively. The curve of Figure 5 (d) and (d') has been shown in the frequency range around 1,500-2,500  $\text{cm}^{-1}$  for the complex of Pb-C\_NC and H@Pb-C\_NC with pointed peaks near 182.37  $\text{cm}^{-1}$  and 1,515.14  $\text{cm}^{-1}$ , respectively.

H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC clusters is approved by the  $\Delta E^\circ_{\text{ads}}$  amounts as formula:

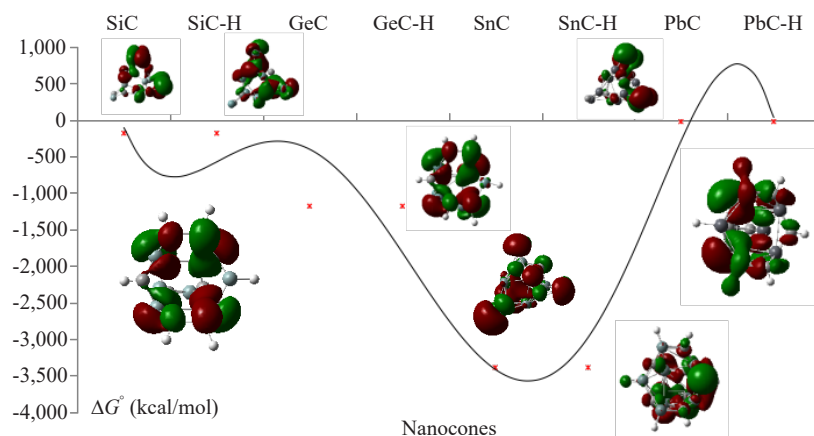
$$\Delta E^\circ_{\text{ads}} = \Delta E^\circ_{\text{H} \rightarrow \text{X-C}_{\text{NC}}} - (\Delta E^\circ_{\text{H}} + \Delta E^\circ_{\text{X-C}_{\text{NC}}}); (X = \text{Si, Ge, Sn, Pb}) \quad (5)$$

Besides, the alterations of  $\Delta H^\circ_{\text{ads}}$  among H-grabbing on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC clusters were measures regarding non-covalent binding extracted from interatomic interactions between H-atoms and nanocarbitides of Si-C\_NC, Ge-C\_NC, Sn-C\_NC and Pb-C\_NC and covalent binding extracted from intra-atomic interactions between C, Si, Ge, Sn, Pb in the nanocone structures (Table 3). As a matter of fact, Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC have greater interaction energy from Van der Waals' forces with H-atoms that can cause them to be much more resistant (Figure 6). The H-grabbing band reflects the saturated hydrogen content on the surface of nanoclusters.

By combining these observed and calculated results, the configurations of hydrogen atoms at the interface of the nanocluster can be discussed.



**Figure 5.** “IR” spectra for (a) Si-C\_NC, (a') H@Si-C\_NC, (b) Ge-C\_NC, (b') H@Ge-C\_NC, (c) Sn-C\_NC, (c') H@Sn-C\_NC, (d) Pb-C\_NC and (d') H@Pb-C\_NC complexes as the selective gas detectors



**Figure 6.** The alterations of Gibbs free energy ( $\Delta G_{\text{ads}}^{\circ}$ ) for H-grabbing onto Si-C\_NC, Ge-C\_NC, Sn-C\_NC and Pb-C\_NC clusters

And  $\Delta G_{\text{ads}}^{\circ}$  is evaluated as the following formula:

$$\Delta G_{\text{ads}}^{\circ} = \Delta G_{\text{H@X-C}_{\text{NC}}}^{\circ} - (\Delta G_{\text{H}}^{\circ} + \Delta G_{\text{X-C}_{\text{NC}}}^{\circ}); \text{ (X = Si, Ge, Sn, Pb)} \quad (6)$$

All the computed amounts of  $\Delta G_{\text{ads}}^{\circ}$  are close (Table 3) that explains the coherence of the measured values by all methods and the validity of the computational methods and force fields (Figure 6).

Table 3 and Figure 6 have displayed that H@Si-C\_NC, H@Ge-C\_NC, H@Sn-C\_NC, H@Pb-C\_NC clusters have the greatest interval of Gibbs free energy adsorption with which describes the alterations between Gibbs free energy of input compounds ( $\Delta G_{\text{H}}^{\circ}$ ) and ( $\Delta G_{\text{Si-C}_{\text{NC}}}^{\circ}$ ,  $\Delta G_{\text{Ge-C}_{\text{NC}}}^{\circ}$ ,  $\Delta G_{\text{Sn-C}_{\text{NC}}}^{\circ}$ ,  $\Delta G_{\text{Pb-C}_{\text{NC}}}^{\circ}$ ) and product compounds ( $\Delta G_{\text{H@Si-C}_{\text{NC}}}^{\circ}$ ,  $\Delta G_{\text{H@Ge-C}_{\text{NC}}}^{\circ}$ ,  $\Delta G_{\text{H@Sn-C}_{\text{NC}}}^{\circ}$ ,  $\Delta G_{\text{H@Pb-C}_{\text{NC}}}^{\circ}$ ) based on the polarizability factor.

Therefore, it is supposed that Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC contain sufficient productivity for H-grabbing on the mentioned nano-carbides through charge transfer from H-atoms to C, Si, Ge, Sn or Pb due to intra-atomic and interatomic interactions.

## 4. Conclusions

In summary, H-grabbing on the nano-carbides of Si-C\_NC, Ge-C\_NC, Sn-C\_NC and Pb-C\_NC was investigated by first-principles calculations. The alterations of charge density illustrated a remarkable charge transfer towards Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC which might play the electron acceptor roles while H-atoms act as the stronger electron donors through adsorption on the Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC. As a matter of fact, Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC have greater interaction energy from Van der Waals' forces with H-atoms that can cause them to be much more resistant. Besides, thermodynamic parameters describing H-grabbing on the nano-carbides of Si-C\_NC, Ge-C\_NC, Sn-C\_NC or Pb-C\_NC have been investigated, including internal process of the adsorbent-adsorbate system. Thermodynamic parameters have constructed a detailed molecular model for atom-atom interactions and a distribution of point charges which can be utilized to reproduce the polarity of the solid material and the adsorbing molecules. Today, it is crucial to distinguish the potential of hydrogen technologies and bring up all perspectives of their performance, from technological advances to economic and social effects. The authors intend to pursue research on sustainability and clean energy subjects towards finding new solutions for reducing the global dependency on fossil fuels.

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## Conflict of interest

The author declares that they have no conflict of interest.

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